# Monolayer Ti<sub>2</sub>CO<sub>2</sub>: A Promising Candidate for NH<sub>3</sub> Sensor or Capturer with High Sensitivity and Selectivity

Xue-fang Yu,<sup>†</sup> Yan-chun Li,<sup>‡</sup> Jian-bo Cheng,<sup>†</sup> Zhen-bo Liu,<sup>†</sup> Qing-zhong Li,<sup>†</sup> Wen-zuo Li,<sup>†</sup> Xin Yang,<sup>†</sup> and Bo Xiao<sup>\*,†</sup>

<sup>†</sup>The Laboratory of Theoretical and Computational Chemistry, School of Chemistry and Chemical Engineering, Yantai University, Yantai 264005, China

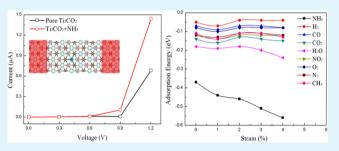
<sup>‡</sup>Institute of Theoretical Chemistry, Jilin University, Changchun 130021, China

#### Supporting Information

ACS APPLIED MATERIALS

XINTERFACES

**ABSTRACT:** Ti<sub>2</sub>C is one of the thinnest layers in MXene family with high potential for applications. In the present study, the adsorption of NH<sub>3</sub>, H<sub>2</sub>, CH<sub>4</sub>, CO, CO<sub>2</sub>, N<sub>2</sub>, NO<sub>2</sub>, and O<sub>2</sub> on monolayer Ti<sub>2</sub>CO<sub>2</sub> was investigated by using first-principles simulations to exploit its potential applications as gas sensor or capturer. Among all the gas molecules, only NH<sub>3</sub> could be chemisorbed on Ti<sub>2</sub>CO<sub>2</sub> with apparent charge transfer of 0.174 e. We further calculated the current–voltage (*I–V*) relation using the nonequilibrium Green's function (NEGF) method. The transport feature exhibits distinct



Research Article

www.acsami.org

responses with a dramatic change of I-V relation before and after NH<sub>3</sub> adsorption on Ti<sub>2</sub>CO<sub>2</sub>. Thus, we predict that Ti<sub>2</sub>CO<sub>2</sub> could be a promising candidate for the NH<sub>3</sub> sensor with high selectivity and sensitivity. On the other hand, the adsorption of NH<sub>3</sub> on Ti<sub>2</sub>CO<sub>2</sub> could be further strengthened with the increase of applied strain on Ti<sub>2</sub>CO<sub>2</sub>, while the adsorption of other gases on Ti<sub>2</sub>CO<sub>2</sub> is still weak under the same strain, indicating that the capture of NH<sub>3</sub> on Ti<sub>2</sub>CO<sub>2</sub> under the strain is highly preferred over other gas molecules. Moreover, the adsorbed NH<sub>3</sub> on Ti<sub>2</sub>CO<sub>2</sub> could be escapable by releasing the applied strain, which indicates the capture process is reversible. Our study widens the application of monolayer Ti<sub>2</sub>CO<sub>2</sub> not only as the battery material, but also as the potential gas sensor or capturer of NH<sub>3</sub> with high sensitivity and selectivity.

KEYWORDS: MXene, monolayer Ti<sub>2</sub>CO<sub>2</sub>, NH<sub>3</sub> sensor, NH<sub>3</sub> capture, sensitivity, selectivity

# 1. INTRODUCTION

 $\rm NH_3$  is the most common substitute for chlorofluorocarbons (CFCs) in cooling systems, and it is a toxic compound and volatile, which is very harmful to the human body.<sup>1</sup> On the other hand,  $\rm NH_3$  has been widely used to synthesize various materials in chemical industries and is a promising candidate as a future energy carrier in vehicles and as an  $\rm NH_3$  reservoir for the selective catalytic reduction (SCR) of  $\rm NO_x$  gases in diesel cars and trucks.<sup>2–6</sup> Thus, it is highly desirable to find an effective method to sensor and capture the  $\rm NH_3$  gases for atmospheric environmental controls and industry applications.

Semiconducting metal oxides and low-dimension materials have been considered to be the most promising candidate for gas sensors, which is due to its advantages such as low cost, small dimensions, and great compatibility with conventional microelectromechanical processing. So far, NH<sub>3</sub> sensors based on metal oxides (such as  $SnO_2$ ,  $TiO_2$ ,  $In_2O_3$ )<sup>7–9</sup> and low-dimension materials (such as carbon nanocone;<sup>10</sup> silicon carbide nanotube;<sup>11</sup> Al, B, Li, N-doped carbon nanotube;<sup>12–15</sup> Ga, Al-doped boron nitride nanotube;<sup>16</sup> and Au, B-doped

graphene<sup>17,18</sup>) have been widely studied. However, the major drawback of these sensors is the lack of selectivity to only one particular gas ( $NH_3$ ), and most of these sensors show long

recovery times at room temperature due to the tendency of NH<sub>3</sub> to strongly interact with many substrates. On the other hand, NH<sub>3</sub> storage has been widely studied, and it is found that the metal ammines  $Mg(NH_3)_6Cl_2$ ,<sup>19</sup> Sr(NH<sub>3</sub>)<sub>8</sub>Cl<sub>2</sub>,<sup>20</sup> and mixed metal halide ammines<sup>21</sup> could store NH<sub>3</sub> very effectively. However, the release of NH<sub>3</sub> from these materials needs either high temperature (>450 K) or multistep reactions, which are not suitable for automotive or fuel cell applications.

Very recently, MXene has attracted worldwide attention due to its structural similarity to graphene.<sup>22,23</sup> MXene could be obtained from the removal of "A" group layer from the MAX phase, where "M" represents an early transition metal, "X" denotes C or N, and "A" is an A-group (mostly IIIA and IVA, or groups 13 and 14) element.<sup>22,23</sup> Since Gogotsi et al. revealed the potential use of MXene layers as the electrochemical energy storage, such as electrode for batteries, supercapacitors, and hybrid devices,<sup>24</sup> several theoretical and experimental studies extensively reported the performance of MXene as anode

```
Received:April 29, 2015Accepted:June 4, 2015Published:June 4, 2015
```

material for metal ion batteries, the spintronic, and the optical devices.  $^{25-28} \!\!$ 

Among all the MXenes,  $Ti_2C$  is one of the thinnest phases, and it is found to be terminated with several kinds of functional group, such as F, OH, and/or O in the real etching process with HF acid.<sup>22,23</sup> It has already been revealed that the electronic property of  $Ti_2C$  is strongly dependent on the functional group, with only  $Ti_2CO_2$  exhibiting semiconductor character,  $Ti_2C$ ,  $Ti_2CF_2$ , and  $Ti_2C(OH)_2$  exhibiting metallic character.<sup>29,30</sup> Thus,  $Ti_2C$  with O termination may possess much more potential applications than other terminations in view of its semiconducting property.<sup>31</sup>

In this work, the potential of monolayer  $Ti_2CO_2$  as the gas sensor and capture is explored by first-principle simulation. The preference adsorption of NH<sub>3</sub> against the other gas molecules (H<sub>2</sub>, CH<sub>4</sub>, CO, CO<sub>2</sub>, N<sub>2</sub>, NO<sub>2</sub> and O<sub>2</sub>) on monolayer  $Ti_2CO_2$ suggests the selectivity of monolayer  $Ti_2CO_2$  toward NH<sub>3</sub>. The distinct transport feature with a dramatic change of *I*–*V* relation before and after NH<sub>3</sub> adsorption indicates the strong sensitivity of monolayer  $Ti_2CO_2$  toward NH<sub>3</sub>. More importantly, the interaction of monolayer  $Ti_2CO_2$  with NH<sub>3</sub> can be tuned easily by strains, indicating the facile way of NH<sub>3</sub> capture and releasing using monolayer  $Ti_2CO_2$ .

### 2. COMPUTATIONAL METHOD

The density functional theory (DFT) calculation was performed using the Vienna Ab-Initio Simulation Package.<sup>32,33</sup> The electron–ion interaction was described by projector augmented-wave (PAW)<sup>34</sup> pseudopotentials. For the exchange and correlation functionals, we use the Perdew–Burke–Ernzerhof (PBE) version of the generalized gradient approximation (GGA).<sup>35</sup> The energy cutoff of 520 eV was used for the wave functions expansion. The Brillouin zone integration was sampled by with  $3 \times 3 \times 1$  *k*-grid mesh for geometry optimization, and  $5 \times 5 \times 1$  *k*-grid mesh for electronic properties calculations to achieve high accuracy. The geometries were fully optimized until the forces on each atom is less than 0.01 eV/Å. The van der Waals interaction is introduced to treat the interaction between gas molecules and the monolayer Ti<sub>2</sub>CO<sub>2</sub>, and it is described by a semiempirical correction by the Grimme method.<sup>36</sup> The periodic structure of Ti<sub>2</sub>CO<sub>2</sub> monolayer (3 × 3 supercell) has been decoupled by a vacuum thickness larger than 20 Å.

The adsorption energy  $(E_{ads})$  is defined as

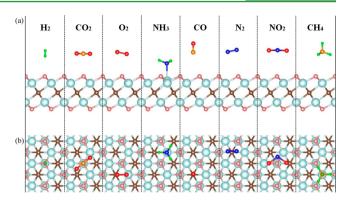
$$E_{ads} = E_{Ti2CO2+Gas} - E_{Gas} - E_{Ti2CO2}$$

where  $E_{\text{Ti2CO2+Gas}}$  is the total energy of monolayer Ti<sub>2</sub>CO<sub>2</sub> adsorbed with gas molecule,  $E_{\text{Gas}}$  is the total energy of gas molecule, and  $E_{\text{Ti2CO2}}$ is the total energy of monolayer Ti<sub>2</sub>CO<sub>2</sub>.

The electron transport calculations were performed using the Atomistix ToolKit (ATK) package<sup>37</sup> based on the density functional theory and nonequilibrium Green's function (NEGF) method. A numerical atomic basis set, a single- $\zeta$  basis with polarization, was used to solve the Kohn–Sham equation. GGA in the PBE form was adopted for the electron–electron interactions. The *K*-point sampling in the direction parallel to the interface (the *x* and *y* directions) is 2 × 2, and the number of real space mesh in the *z* direction is 50.

# 3. RESULTS AND DISCUSSION

The most stable structure of O functionalized monolayer Ti<sub>2</sub>C, as mentioned in our previous study,<sup>28</sup> is shown in Figure 1. The lattice parameter of Ti<sub>2</sub>CO<sub>2</sub> unit cell is 3.04 Å, which is consistent with other theoretical studies.<sup>28,29</sup> We first consider several typical and possible adsorption sites for the gas molecules (NH<sub>3</sub>, H<sub>2</sub>, CH<sub>4</sub>, CO, CO<sub>2</sub>, N<sub>2</sub>, NO<sub>2</sub>, and O<sub>2</sub>) adsorption on the monolayer Ti<sub>2</sub>CO<sub>2</sub>, including the top site over Ti, C, or O atom, bridge site between neighboring O and



**Research Article** 

Figure 1. A schematic illustration of (a) side, and (b) top view of the adsorption of  $NH_{3}$ ,  $H_2$ ,  $CH_4$ , CO,  $CO_2$ ,  $N_2$ ,  $NO_2$ , or  $O_2$  molecule on monolayer  $Ti_2CO_2$ .

O, or Ti and O atoms. Also, several typical orientation of gas molecules with respective to the monolayer  $\rm Ti_2CO_2$  surface are considered. Taking NH<sub>3</sub> molecule's adsorption as an example, NH<sub>3</sub> molecule is initially placed vertically (including Ti<sub>2</sub>CO<sub>2</sub>  $\leftrightarrow$  H–NH<sub>2</sub>, and Ti<sub>2</sub>CO<sub>2</sub>  $\leftrightarrow$  H<sub>2</sub>–NH) or parallel (including Ti<sub>2</sub>CO<sub>2</sub>  $\leftrightarrow$  N–H<sub>3</sub>, and Ti<sub>2</sub>CO<sub>2</sub>  $\leftrightarrow$  H<sub>3</sub>–N) to the surface of monolayer Ti<sub>2</sub>CO<sub>2</sub> for all the studied adsorption sites, and in each case, all the typical orientation of NH<sub>3</sub> molecules are considered.

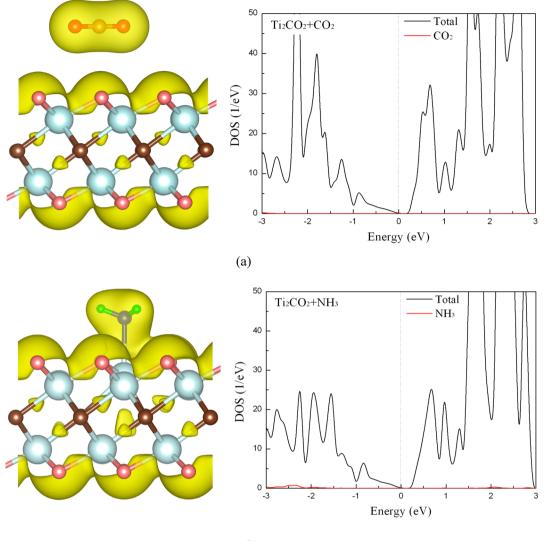
The most stable structures for the adsorption of gas molecules on monolayer  $Ti_2CO_2$  are shown in Figure 1, and the corresponding adsorption energies, charge transfers, and the nearest distance between gas molecules and  $Ti_2CO_2$  sheet (denote as  $d_{Gas-T2CO_2}$ ) are listed in Table 1. In the case of NH<sub>3</sub>

Table 1. Adsorption Energy  $(E_{ads})$ , the Charge Transfer  $(C_T)$  from Gas Molecule to Ti<sub>2</sub>CO<sub>2</sub>, and the Nearest Distance  $(d_{Gas-Ti2CO2})$  between Gas Molecule and Ti<sub>2</sub>CO<sub>2</sub>

	gas molecules	$E_{\rm ads}$ (eV)	$C_{\rm T}~({\rm e})^a$	$d_{\text{Gas-Ti2CO2}}$ (Å)
	NH <sub>3</sub>	-0.37	0.174	2.35
	$H_2$	-0.05	0.003	2.76
	CO	-0.07	0.004	3.19
	$CO_2$	-0.14	-0.007	3.08
	NO <sub>2</sub>	-0.12	-0.003	3.35
	O <sub>2</sub>	-0.08	-0.003	3.46
	$N_2$	-0.12	-0.002	3.71
	$CH_4$	-0.11	-0.001	3.06
a A	nositivo C	value indicator e	transfor of alact	rong from the go

 $^aA$  positive  $C_T$  value indicates a transfer of electrons from the gas molecule to  ${\rm Ti}_2{\rm CO}_2.$ 

molecule, the strongest binding site is found to be the former with the N atom located directly above the Ti atom in  $Ti_2CO_2$ , as shown in Figure 1. This structure has the adsorption energy of -0.37 eV and a charge transfer of 0.174 e from the NH<sub>3</sub> to the  $Ti_2CO_2$ . The as-formed N-Ti bond length is 2.35 Å. Furthermore, the NH<sub>3</sub> adsorption induces a locally structural deformation to both NH<sub>3</sub> molecule and  $Ti_2CO_2$ . The dihedral angle of H-H-H-N in NH<sub>3</sub> decreases from 39.50° in free NH<sub>3</sub> to 35.42° in the adsorbed form. The NH<sub>3</sub>-adsorbed Ti atom is pulled outward from the layer with its neighboring Ti-C bond length increase from 2.08 to 2.35 Å. The three O atoms adjacent to N atom have slightly moved away from its original position due to the repulsion between O atoms and N atoms. The local structure deformation of NH<sub>3</sub> with monolayer



(b)

Figure 2. Total charge density and the density of states of (a) CO2 and (b) NH3 adsorption on monolayer Ti2CO2.

 $Ti_2CO_2$ , which is very consistent with the small adsorption energy.

In contrast, other gas molecules (H<sub>2</sub>, CH<sub>4</sub>, CO, CO<sub>2</sub>, N<sub>2</sub>,  $NO_2$ , and  $O_2$ ) exhibit quite different adsorption behavior compared with the case of NH<sub>3</sub>. As shown in Table 1, the adsorption energies of these gas molecules on monolayer  $Ti_2CO_2$  range from -0.05 to -0.14 eV, and the charge transfers between gas molecules and monolayer Ti<sub>2</sub>CO<sub>2</sub> are from -0.007 to 0.004 e, which are obviously smaller than that in the case of  $NH_3$  (adsorption energy -0.37 eV, and charge transfer 0.174 e). As shown in Figure 1, the adsorption of these gas molecules has a relatively large  $d_{\text{Gas}-\text{T2CO2}}$  value and the change in the structures of gas molecules and Ti<sub>2</sub>CO<sub>2</sub> is negligible, which implies the weak interaction between them. In detail, as shown in Figure 1,  $H_2$  (or CO) molecule tends to be vertically adsorbed on the O (or C) site of Ti<sub>2</sub>CO<sub>2</sub>. The  $d_{H2(or CO)-T2CO2}$ and the adsorption energy is 2.76 (or 3.19) Å and -0.05 (or -0.07) eV, respectively; CO<sub>2</sub> (or NO<sub>2</sub>) molecule prefers to adsorb on the O site of  $Ti_2CO_2$  and the O-C-O (or O-N-O) bond is parallel to the  $Ti_2CO_2$  sheet, the  $d_{CO2(or NO2)-T2CO2}$ and the adsorption energy are 3.08 (or 3.35) Å and -0.14 (or -0.12) eV, respectively; O<sub>2</sub> (or N<sub>2</sub>) molecule prefers to adsorb

on the O site of Ti<sub>2</sub>CO<sub>2</sub> sheet with a tilt structure, the  $d_{O2(or N2)-T2CO2}$  and the adsorption energy are 3.46 (or 3.70) Å and -0.08 (or -0.12) eV, respectively. Lastly, CH<sub>4</sub> molecule tends to be adsorbed on the O-site with one C–H bond perpendicular to the surface of Ti<sub>2</sub>CO<sub>2</sub> sheet while the other three C–H bonds point to the O atoms. The calculated  $d_{CH4-T2CO2}$  and the adsorption energy are 3.06 Å and -0.11 eV, respectively.

To analyze the electronic interaction between the gas molecules and  $Ti_2CO_2$  sheet, we calculate the density of state (DOS) and total charge density. Taking the adsorption of NH<sub>3</sub> and CO<sub>2</sub> on  $Ti_2CO_2$  as examples, in the case of CO<sub>2</sub>, the total charge density (Figure 2a) shows no overlap between CO<sub>2</sub> and  $Ti_2CO_2$ , indicating the weak interaction between them, and correspondingly the change in the DOS of  $Ti_2CO_2$  is negligible as shown in Figure 2a. Similar results also could be found in the case of the other gas molecules (including H<sub>2</sub>, CH<sub>4</sub>, CO, N<sub>2</sub>, NO<sub>2</sub>, and O<sub>2</sub>). In view of the weak interaction and the few charge transfer between  $Ti_2CO_2$  and these gas molecules (including H<sub>2</sub>, CH<sub>4</sub>, CO, CO<sub>2</sub>, N<sub>2</sub>, NO<sub>2</sub>, and O<sub>2</sub>),  $Ti_2CO_2$  could not be the sensor to these gas molecules. In contrast, in the case of NH<sub>3</sub>, the electronic charge are strongly overlapped,

### **ACS Applied Materials & Interfaces**

as shown in Figure 2b, leading to orbital mixing and large charge transfer. The adsorption of NH<sub>3</sub> induces several distinct states at the lower-lying valence bands in the energy range around -2.5 eV; however, there is no noticeable modifications of the DOS near Fermi level ( $E_{\rm F}$ ), as shown in Figure 2b. Thus, the adsorption of NH<sub>3</sub> does not have a substantial effect on the electronic structure of monolayer Ti<sub>2</sub>CO<sub>2</sub>. It should be noted that the adsorption-induced charge transfer is expected to affect the resistivity of the system, which can be measured experimentally and could be a marker for gas sensors. The evidence could be found in detecting NH<sub>3</sub> molecule by using MoS<sub>2</sub> and phosphorene,<sup>38,39</sup> where the resistivity of the system increases obviously after the adsorption of NH<sub>3</sub> due to the relatively large charge transfer between them, although the change in the electronic properties of system is negligible.

To quantitatively evaluate the performance of monolayer  $Ti_2CO_2$  as the NH<sub>3</sub> sensor, we simulated the current–voltage (I-V) relations before and after the NH<sub>3</sub> adsorption using the NEGF method, which allows the monitoring of the resistivity change in sensing materials upon the adsorption of chemicals.  $Ti_2CO_2$ -based sensor is simulated using a model consisting of a  $Ti_2CO_2$  sheet contacted by two  $Ti_2CO_2$  electrodes, as shown in Figure 3a. The I-V curves for such  $Ti_2CO_2$  junctions with and

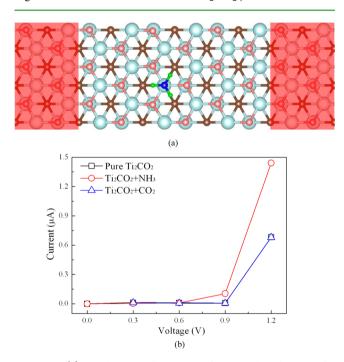


Figure 3. (a) A schematic illustration of  $Ti_2CO_2$ -based sensor for detecting  $NH_3$  molecule; (b) the current–voltage (*I–V*) relations before and after the adsorption of  $NH_3$  or  $CO_2$  molecule on monolayer  $Ti_2CO_2$ .

without the adsorption of NH<sub>3</sub> molecule is shown in Figure 3b. The I-V curves of Ti<sub>2</sub>CO<sub>2</sub> exhibits a clear nonlinear behavior. After the adsorption of NH<sub>3</sub> molecule, a dramatic increase of current is observed when the applied bias voltage larger than 0.9 V, indicating the high sensitivity of Ti<sub>2</sub>CO<sub>2</sub> sensor to NH<sub>3</sub>. In detail, the current passing through the monolayer Ti<sub>2</sub>CO<sub>2</sub> is 0.68  $\mu$ A under the bias voltage of 1.2 V, but with the adsorption of NH<sub>3</sub> molecule, the current under the same bias is increased to 1.44  $\mu$ A, which is about the 111% increase. The increase in the current indicates the decrease of resistance after the NH<sub>3</sub> adsorption, which could be directly measured in experiment. As a comparison, the I-V curves before and after the adsorption of CO<sub>2</sub> molecule on Ti<sub>2</sub>CO<sub>2</sub> have been calculated as shown in Figure 3b. In this case, the current change is negligible with the increase of applied bias voltage, and the main reason may result from the little charge transfer (-0.007 e) between them. Similar I-V character is also expected to occur when the other gas molecules (i.e., H<sub>2</sub>, CH<sub>4</sub>, CO, N<sub>2</sub>, NO<sub>2</sub> and O<sub>2</sub>) adsorb on Ti<sub>2</sub>CO<sub>2</sub> due to the same reason. This result further confirms the high selectivity of Ti<sub>2</sub>CO<sub>2</sub> as the NH<sub>3</sub> sensor. In view of the dramatically increased current value after the NH<sub>3</sub> adsorption on Ti<sub>2</sub>CO<sub>2</sub>, together with the relatively large charge transfer (0.174 e) between NH<sub>3</sub> and Ti<sub>2</sub>CO<sub>2</sub> as compared with other gas molecules (from -0.007 to 0.004 e), we predict that monolayer Ti<sub>2</sub>CO<sub>2</sub> is a promising candidate to sensor NH<sub>3</sub> molecule with high sensitivity and selectivity. It should be noted that the adsorption energy of NH<sub>3</sub> on  $Ti_2CO_2$  is -0.37 eV, which is suitable for adsorption/desorption of gases on/from Ti<sub>2</sub>CO<sub>2</sub> surface, and thus, the Ti<sub>2</sub>CO<sub>2</sub> sensor could recover easily after detecting NH<sub>3</sub>.

As mentioned above,  $MoS_2$  and phosphorene have been proposed to be the NH<sub>3</sub> sensors.<sup>38,39</sup> In this work, we also theoretically estimate the performance of  $MoS_2$  and phosphorene as the NH<sub>3</sub> sensor in comparison with Ti<sub>2</sub>CO<sub>2</sub>. To do so, we have initially considered all the possible structures for the adsorption of NH<sub>3</sub> on  $MoS_2$  and phosphorene, and the corresponding most stable structures are very similar to the previous studies.<sup>38,40</sup> The calculated adsorption energies and the charge transfers between NH<sub>3</sub> and  $MoS_2$  (and phosphorene) are listed in Table 2. These results are very consistent with

Table 2. Adsorption Energy  $(E_{ads})$  and Charge Transfer  $(C_T)$  from NH<sub>3</sub> Molecule to Different 2D Materials

	gas molecules	$E_{\rm ads}$ (eV)	$C_{\mathrm{T}}$ (e) <sup><i>a</i></sup>
	Ti <sub>2</sub> CO <sub>2</sub>	-0.37	0.174
	$Ti_2C(OH)_2$	-0.48	-0.117
	$Ti_3C_2O_2$	-0.34	0.166
	$V_2CO_2$	-0.81	0.265
	phosphorene	-0.21	0.028
	MoS <sub>2</sub>	-0.16	0.040
7.			

 $^{a}A$  positive  $C_{\rm T}$  value indicates a transfer of electrons from the NH<sub>3</sub> molecule to 2D materials.

previous studies.<sup>38,40</sup> Similar to the case of Ti<sub>2</sub>CO<sub>2</sub>, the electronic properties of MoS<sub>2</sub> and phosphorene after the adsorption of NH<sub>3</sub> are not substantially changed,<sup>38,40</sup> and thus, the adsorption-induced charge transfer could act as a marker for gas sensors. It is found that the charge transfer from NH<sub>3</sub> to  $MoS_2$  (0.040 e) or phosphorene (0.028 e) is obviously smaller than that in the case of  $Ti_2CO_2$  (0.174 e). Correspondingly, the calculated I-V curve (Figure 4) before and after the adsorption of NH<sub>3</sub> on MoS<sub>2</sub> or phosphorene shows less current change as compared with the case in  $Ti_2CO_2$ , which implies the superior sensitivity of Ti2CO2 as the NH3 sensor as compared with  $MoS_2$  and phosphorene. It should be noted that the adsorption energy of NH<sub>3</sub> on MoS<sub>2</sub> or phosphorene is smaller than that of Ti<sub>2</sub>CO<sub>2</sub>, and thus, shorter recovery time is needed in MoS<sub>2</sub> or phosphorene compared with the one in Ti<sub>2</sub>CO<sub>2</sub> after detecting NH<sub>3</sub>.

As we know,  $Ti_2CO_2$  is one of the members among MXene family.<sup>22,23</sup> It would be very helpful to learn about the adsorption of NH<sub>3</sub> molecule on other kinds of MXenes with different surface terminations (i.e.,  $Ti_2C(OH)_2$ ), structures (i.e.,

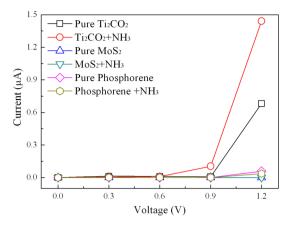
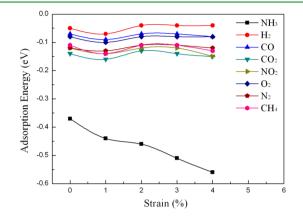


Figure 4. Calculated I-V curve before and after the adsorption of NH<sub>3</sub> on Ti<sub>2</sub>CO<sub>2</sub>, MoS<sub>2</sub>, and phosphorene.

 $Ti_3C_2O_2$ ), and transition metals (i.e.,  $V_2CO_2$ ). To do so, we have considered all the possible structures for adsorption of NH<sub>3</sub> molecule on monolayer Ti<sub>2</sub>C(OH)<sub>2</sub>, Ti<sub>3</sub>C<sub>2</sub>O<sub>2</sub> and V<sub>2</sub>CO<sub>2</sub>, and the corresponding most stable structures are shown in Figure S1 in the Supporting Information. For the adsorption of  $NH_3$  on  $Ti_2C(OH)_2$ , the N atom of  $NH_3$  is located directly on the top of H atom in  $Ti_2C(OH)_2$  in the most stable structure, while for the adsorption of NH<sub>3</sub> on Ti<sub>3</sub>C<sub>2</sub>O<sub>2</sub> or V<sub>2</sub>CO<sub>2</sub>, the most stable structure is similar to the case in  $Ti_2CO_2$ , where the N atom of NH<sub>3</sub> adsorb on the top of transition metal (Ti or V). The corresponding adsorption energy and charge transfer values are listed in Table 2. It is found that the absolute values of charge transfer in all the above cases are larger than 0.117 e, which indicates the strong interaction of NH<sub>3</sub> with them. As mentioned above, the large charge transfer values could be measured experimentally and could be a marker for gas sensors. While the larger adsorption energy of NH<sub>3</sub> on  $Ti_2C(OH)_2$ (-0.48 eV) or V<sub>2</sub>CO<sub>2</sub> (-0.81 eV) implies the longer recover time after detecting NH<sub>3</sub>, compared with NH<sub>3</sub> absorption on  $Ti_2CO_2$ . In the case of NH<sub>3</sub> adsorption on  $Ti_3C_2O_2$ , due to the surface structural similarity between Ti<sub>2</sub>CO<sub>2</sub> and Ti<sub>3</sub>C<sub>2</sub>O<sub>2</sub>, both the adsorption energy and the charge transfer between NH<sub>3</sub> and  $Ti_3C_2O_2$  are very close to the ones in NH<sub>3</sub> absorption on  $Ti_2CO_2$ , and thus, it is expected that the  $Ti_3C_2O_2$  could be another promising candidate for NH<sub>3</sub> sensor.

Next, we consider the possibility of Ti<sub>2</sub>CO<sub>2</sub> as the capturer for NH<sub>3</sub> gas. As mentioned above, the adsorption energy of NH<sub>3</sub> on monolayer Ti<sub>2</sub>CO<sub>2</sub> is -0.37 eV. According to Guo et al., the adsorption energy should be greater than -0.50 eV to effectively capture gas molecule on solid surface.<sup>41</sup> Thus, the perfect monolayer Ti<sub>2</sub>CO<sub>2</sub> is unsuitable for NH<sub>3</sub> gas capture due to its relatively small adsorption energy (-0.37 eV). As shown in Figure 2b, after the adsorption of NH<sub>3</sub> on Ti<sub>2</sub>CO<sub>2</sub>, the charge transfer mainly occurs between the N atom of NH<sub>3</sub> and Ti atom of Ti<sub>2</sub>CO<sub>2</sub>, and thus, the modification of charge state on N or Ti atom might result in the change of adsorption energy. On the other hand, our previous study showed that the charge state on Ti atom of Ti<sub>2</sub>CO<sub>2</sub> decreases with the increase of the applied strains.<sup>28</sup> Thus, we assume that more electrons might transfer from the N of NH<sub>3</sub> to the Ti of Ti<sub>2</sub>CO<sub>2</sub> after NH<sub>3</sub> adsorption to compensate the decreased electrons on the Ti atom of monolayer  $Ti_2CO_2$  if strain is applied; consequently, the adsorption of NH<sub>3</sub> molecule on Ti<sub>2</sub>CO<sub>2</sub> might be further strengthened after applying strains.

To verify this ideal, the adsorption of gas molecules on monolayer  $Ti_2CO_2$  with biaxial strains (from 1 to 4%) have been studied, where the strain is defined as  $\varepsilon = (a - a_0)/a_0$ , where  $a_0$  and a are the lattice parameter of the unit cell without strain and strain, respectively. As shown in Figure 5, the



**Figure 5.** Adsorption energies of gas molecules (including  $NH_3$ ,  $H_2$ ,  $CH_4$ , CO,  $CO_2$ ,  $N_2$ ,  $NO_2$ , or  $O_2$ ) on monolayer  $Ti_2CO_2$  as a function of applied biaxial strains (from 0 to 4%).

adsorption energy of NH<sub>3</sub> on Ti<sub>2</sub>CO<sub>2</sub> dramatically decreases with the increase of applied biaxial strains on Ti<sub>2</sub>CO<sub>2</sub>. For example, the adsorption energy becomes -0.51 eV when 3% biaxial strain is applied. Taking the adsorption energy of -0.50eV as the reference<sup>41</sup> for the gas capturer, our results reveal that the NH<sub>3</sub> gas could be effectively captured on monolayer  $Ti_2CO_2$  with the biaxial strains larger than 3%. In contrast, the adsorption energies of other gas molecules on Ti<sub>2</sub>CO<sub>2</sub> only change slightly after applying biaxial strains, and the interaction between them is still very weak, indicating that the capture of NH<sub>3</sub> on monolayer Ti<sub>2</sub>CO<sub>2</sub> under the biaxial strain is highly preferred over other gas molecules. Moreover, the capture of NH<sub>3</sub> on monolayer Ti<sub>2</sub>CO<sub>2</sub> is reversible, that is, the adsorbed NH<sub>3</sub> can be released by releasing the applied biaxial strains. Thus, we predict that the monolayer Ti<sub>2</sub>CO<sub>2</sub> could be a promising material for efficient NH3 capture, separation, and storage. It should be noted that the sensitive strain dependence behavior also has been reported in the adsorption of NH<sub>3</sub> molecules on monolayer MoS<sub>2</sub>.<sup>42</sup>

Finally, we present the simulated X-ray diffraction (XRD) patterns and constant-current scanning tunneling microscopy (STM) for the Ti<sub>2</sub>CO<sub>2</sub> structure before and after NH<sub>3</sub> adsorption, as shown in Figure S2 (Supporting Information). In XRD patterns (Figure S2a, Supporting Information), the first and second strong peaks are located between the diffraction angle 0 and 10°, and the intensity of the second peak decrease slightly after the NH<sub>3</sub> adsorption. In Figure S2b (Supporting Information), the STM images are simulated around the Fermi level  $E_{\rm F}$  (ranging from  $E_{\rm F}$  –0.5 eV to  $E_{\rm F}$ ), employing the Tersoff–Hamann approach.<sup>43</sup> It is found that the bright protrusion appears after the adsorption of NH<sub>3</sub> on Ti<sub>2</sub>CO<sub>2</sub>, which is contributed from NH<sub>3</sub> molecule. All these results can provide useful information for future experimental studies for identifying the structure from the experimental samples.

# 4. CONCLUSION

By using first-principle simulation, we show that  $Ti_2CO_2$  could be a promising candidate for the NH<sub>3</sub> sensor with high selectivity and sensitivity, which is deduced on the basis of the following evidence: (1) only  $NH_3$  could be chemisorbed on the monolayer  $Ti_2CO_2$  as compared with other gas molecules ( $H_2$ ,  $CH_4$ , CO, CO<sub>2</sub>, N<sub>2</sub>, NO<sub>2</sub>, and O<sub>2</sub>); (2) the adsorption energy of NH<sub>3</sub> on Ti<sub>2</sub>CO<sub>2</sub> is -0.37 eV, which is suitable for adsorption/desorption of gases on/from solid surface, and thus, the Ti<sub>2</sub>CO<sub>2</sub> sensor could recover easily after detecting NH<sub>3</sub>; (3) the electronic conductivity of Ti<sub>2</sub>CO<sub>2</sub> has been significantly enhanced after the adsorption of NH<sub>3</sub>, leading to the high sensitivity of the Ti<sub>2</sub>CO<sub>2</sub> sensor to NH<sub>3</sub>. In addition, the interaction between NH<sub>3</sub> and Ti<sub>2</sub>CO<sub>2</sub> could be further enhanced by applying strains on Ti<sub>2</sub>CO<sub>2</sub>, that is, the adsorption energy becomes -0.51 eV when 3% strain is applied, while the adsorption of other gases on Ti<sub>2</sub>CO<sub>2</sub> is much weaker than that of NH<sub>3</sub> under the same strain. Thus, our results show that Ti<sub>2</sub>CO<sub>2</sub> could be used as the NH<sub>3</sub> capture, separation, and storage material by controlling strains.

# ASSOCIATED CONTENT

# **S** Supporting Information

Structure and total charge density for the adsorption of  $NH_3$  on  $Ti_2C(OH)_2$ ,  $Ti_3C_2O_2$ , and  $V_2CO_2$  sheet; simulated XRD pattern and STM for the  $Ti_2CO_2$  structure before and after  $NH_3$  adsorption. The Supporting Information is available free of charge on the ACS Publications website at DOI: 10.1021/ acsami.5b03737.

#### AUTHOR INFORMATION

### **Corresponding Author**

\*E-mail: xiaoboy8@gmail.com.

#### Notes

The authors declare no competing financial interest.

# ACKNOWLEDGMENTS

This work was partially supported by the Natural Science Foundation of Shandong Province (no. ZR2013BM016). A part of the computation in this work has been done using the Supercomputing Environment of Chinese Academy of Sciences.

### REFERENCES

(1) Licyayo, D. C. M.; Suzuki, A.; Matsumoto, M. Interactions Among Ammonia Fungi on MY Agar Medium with Varying pH. *Mycoscience* **2007**, *48*, 20–28.

(2) Christensen, C. H.; Johannessen, T.; Sorensen, R. Z.; Norskov, J. K. Towards An Ammonia-Mediated Hydrogen Economy? *Catal. Today* **2006**, *111*, 140–144.

(3) Chakraborty, D.; Petersen, H. N.; Elkjaer, C.; Cagulada, A.; Johannessen, T. Solid Ammonia As Energy Carrier: Current Status and Future Prospects. *Fuel Cells Bull.* **2009**, 2009, 12–15.

(4) Lan, R.; Irvine, J. T. S.; Tao, S. Ammonia and Related Chemicals As Potential Indirect Hydrogen Storage Materials. *Int. J. Hydrogen Energy* **2012**, *37*, 1482–1494.

(5) Schuth, F.; Palkovits, R.; Schlogl, R.; Su, D. S. Ammonia as a Possible Element in an Energy Infrastructure: Catalysts for Ammonia Decomposition. *Energy Environ. Sci.* **2012**, *5*, 6278–6289.

(6) Elmoe, T. D.; Sorensen, R. Z.; Quaade, U.; Christensen, C. H.; Norskov, J. K.; Johannessen, T. A High-Density Ammonia Storage/ Delivery System Based on  $Mg(NH_3)_6Cl_2$  for SCR-DeNO<sub>x</sub> in Vehicles. *Chem. Eng. Sci.* **2006**, *61*, 2618–2625.

(7) Wang, Y. D.; Wu, X. H.; Su, Q.; Li, Y. F.; Zhou, Z. L. Ammonia-Sensing Characteristics of Pt and SiO<sub>2</sub> Doped SnO<sub>2</sub> Materials. *Solid-State Electron.* **2011**, *45*, 347–350. (8) Shimizu, Y.; Okamato, T.; Takao, Y.; Egashira, M. Desorption Behavior of Ammonia from TiO<sub>2</sub> Based Specimens-Ammonia Sensing Mechanism of Double Layer Sensors with TiO<sub>2</sub> Based Catalyst Layers. *J. Mol. Catal.* **2000**, *155*, 183–191.

(9) Guo, P.; Pan, H. Selectivity of Ti-Doped  $In_2O_3$  Ceramics as An Ammonia Sensor. Sens. Actuators, B **2006**, 114, 762–767.

(10) Baei, M. T.; Peyghan, A. A.; Bagheri, Z. Carbon Nanocone as an Ammonia Sensor: DFT Studies. *Struct Chem.* **2013**, *24*, 1099–1103.

(11) Ganji, M. D.; Seyed-Aghaei, N.; Taghavi, M. M.; Rezvani, M.; Kazempour, F. Ammonia Adsorption on SiC Nanotubes: A Density Functional Theory Investigation. *Fullerenes, Nanotubes, Carbon Nanostruct.* **2011**, *19*, 289–299.

(12) Azizi, K.; Karimpanah, M. Computational Study of Al- or P-doped Single-Walled Carbon Nanotubes as  $NH_3$  and  $NO_2$  Sensors. *Appl. Surf. Sci.* **2013**, 285P, 102–109.

(13) Bai, L.; Zhou, Z. Computational Study of B- or N-Doped Single-Walled Carbon Nanotubes as  $NH_3$  and  $NO_2$  Sensors. *Carbon.* 2007, 45, 2105–2110.

(14) Goudarzi, F.; Vaezi, M. R.; Kazemzadeh, A. A Novel Single Wall Carbon Nanotubes-Based Sensor Doped with Lithium for Ammonia Gas Detection. J. Ceram. Process. Res. **2012**, 13, 612–616.

(15) Battie, Y.; Ducloux, O.; Thobois, P.; Susi, T.; Kauppinen, E. I.; Loiseau, A. Selective Differential Ammonia Gas Sensor Based On N-Doped SWCNT Films. *Phys. Status Solidi B* **2011**, *248*, 2462–2466.

(16) Soltani, A.; Raz, S. G.; Rezaei, V. J.; Khalaji, A. D.; Savar, M. Ab Initio Investigation of Al- and Ga-Doped Single-Walled Boron Nitride Nanotubes as Ammonis Sensor. *Appl. Surf. Sci.* **2012**, *263*, 619–625.

(17) Gautam, M.; Jayatissa, A. H. Ammonia Gas Sensing Behavior of Graphene Surface Decorated with Gold Nanoparticles. *Solid-State Electron.* **2012**, *78*, 159–165.

(18) Zhang, Y. H.; Chen, Y. B.; Zhou, K. G.; Liu, C. H.; Zeng, J.; Zhang, H. L.; Peng, Y. Improving Gas Sensing Properties of Graphene by Introducing Dopants and Defects: A First-Principles Study. *Nanotechnology*. **2009**, *20*, 185504–185511.

(19) Sorensen, R. Z.; Hummelshoj, J. S.; Klerke, A.; Reves, J. B.; Vegge, T.; Norskov, J. K.; Christensen, C. H.; Sorensen, R. Z.; Hummelshoj, J. S.; Norskov, J. K. Indirect, Reversible High-Density Hydrogen Storage in Compact Metal Ammine Salts. *J. Am. Chem. Soc.* **2008**, *130*, 8660–8668.

(20) Lysgaard, S.; Ammitzboll, A. L.; Johnsen, R. E.; Norby, P.; Quaade, U. J.; Vegge, T. Resolving the Stability and Structure of Strontium Chloride Amines from Equilibrium Pressures, XRD and DFT. *Int. J. Hydrogen Energy.* **2012**, *37*, 18927–18936.

(21) Jensen, P. B.; Lysgaard, S.; Quaade, U. J.; Vegge, T. Designing Mixed Metal Halide Ammines for Ammonia Storage Using Density Functional Theory And Genetic Algorithms. *Phys. Chem. Chem. Phys.* **2014**, *16*, 19762–19740.

(22) Naguib, M.; Mochalin, V. N.; Barsoum, M. W.; Gogotsi, Y. 25th Anniversary Article: MXenes: A New Family of Two-Dimensional Materials. *Adv. Mater.* **2014**, *26*, 992–1005.

(23) Naguib, M.; Gogotsi, Y. Synthesis of Two-Dimensional Materials by Selective Extraction. *Acc. Chem. Res.* **2015**, *48*, 128–135. (24) Lukatskaya, M. R.; Mashtalir, O.; Ren, C. E.; Dall'Agnese, Y.; Rozier, P.; Taberna, P. L.; Naguib, M.; Simon, P.; Barsoum, M. W.; Gogotsi, Y. Cation Intercalation and High Volumetric Capacitance of Two-Dimensional Titanium Carbide. *Science* **2013**, *341*, 1502–1505.

(25) Yu, X. F.; Cheng, J. B.; Liu, Z. B.; Li, Q. Z.; Li, W. Z.; Yang, X.; Xiao, B. Mg Intercalation into  $Ti_2C$  Building Block. *Chem. Phys. Lett.* **2015**, 629, 36–39.

(26) Xie, Y.; Naguib, M.; Mochalin, V. N.; Barsoum, M. W.; Gogotsi, Y.; Yu, X.; Nam, K.-W.; Yang, X.-Q.; Kolesnikov, A. I.; Kent, P. R. Role of Surface Structure on Li-Ion Energy Storage Capacity of Two Dimensional Transition Metal Carbides. J. Am. Chem. Soc. 2014, 136, 6385–6394.

(27) Peng, X.; Peng, L.; Wu, C.; Xie, Y. Two Dimensional Nanomaterials for Flexible Supercapacitors. *Chem. Soc. Rev.* 2014, 43, 3303–3323.

# **ACS Applied Materials & Interfaces**

(28) Yu, X. F.; Cheng, J. B.; Liu, Z. B.; Li, Q. Z.; Li, W. Z.; Yang, X.; Xiao, B. The Band Gap Modulation of Monolayer  $Ti_2CO_2$  by Strain. *RSC Adv.* **2015**, *5*, 30438–30444.

(29) Khazaei, M.; Arai, M.; Sasaki, T.; Chung, C.-Y.; Venkataramanan, N. S.; Estili, M.; Sakka, Y.; Kawazoe, Y. Novel Electronic and Magnetic Properties of Two-Dimensional Transition Metal Carbides and Nitrides. *Adv. Funct. Mater.* **2013**, *23*, 2185–2192.

(30) Kurtoglu, M.; Naguib, M.; Gogotsi, Y.; W.Barsoum, M. First Principles Study of Two-Dimensional Early Transition Metal Carbides. *MRS Commun.* **2012**, *2*, 133–137.

(31) Come, J.; Naguib, M.; Rozier, P.; Barsoum, M. W.; Gogotsi, Y.; Taberna, P. L.; Morcrette, M.; Simon, P. A Non-Aqueous Asymmetric Cell with a Ti<sub>2</sub>C-Based Two-Dimensional Negative Electrode. *J. Electrochem. Soc.* **2012**, *159*, A1368–A1373.

(32) Kresse, G.; Furthmüller, J. Efficiency of ab-Initio Total Energy Calculations for Metals and Semiconductors Using a Plane-Wave Basis Set. *Comput. Mater. Sci.* **1996**, *6*, 15–50.

(33) Kresse, G.; Furthmüller, J. Efficient Iterative Schemes for ab-Initio Total-Energy Calculations Using a Plane-Wave Basis Set. *Phys. Rev. B* **1996**, *54*, 11169–11186.

(34) Kresse, G.; Joubert, J. From Ultrasoft Pseudopotentials to the Projector Augmented-Wave Method. *Phys. Rev. B* **1999**, *59*, 1758–1775.

(35) Wang, Y.; Perdew, J. P. Correlation Hole of the Spin-Polarized Electron Gas, with Exact Small-Wave-Vector and High-Density Scaling. *Phys. Rev. B* **1991**, *44*, 13298–13307.

(36) Grimme, S. Semiempirical GGA-type Density Functional Constructed with A Long-Range Dispersion Correction. *J. Comput. Chem.* **2006**, *30*, 1787–1799.

(37) Brandbyge, M.; Mozos, J. L.; Ordejon, P.; Taylor, J.; Stokbro, K. Density-Functional Method for Nonequilibrium Electron Transport. *Phys. Rev. B* **2002**, *65*, 165401.

(38) Cho, B.; Hahm, M. G.; Choi, M.; Yoon, J.; Kim, A. R.; Lee, Y. J.; Park, S. G.; Kwon, J. D.; Kim, C. S.; Song, M. K.; J, Y.; Nam, K. S.; Lee, S.; Yoo, T. J.; Kang, C. C.; Lee, B. H.; Ko, H. C.; Ajayan, P. M.; Kim, D. H. Charge-Transfer-Based Gas Sensing Using Atomic-Layer  $MOS_2$ . *Sci. Rep.* **2015**, *5*, 8052.

(39) Kou, L. Z.; Frauenheim, T.; Chen, C. F. Phosphorene as A Superior Gas Sensor: Selective Adsorption and Distince I-V Response. J. Phys. Chem. Lett. **2014**, 5, 2675–2681.

(40) Cai, Y. Q.; Ke, Q. Q.; Zhang, G.; Zhang, Y. W. Energetics, Charge Transfer and Magnetism of Small Molecules Physisorbed on Phosphorene. J. Phys. Chem. C 2015, 119, 3102–3110.

(41) Guo, H. Y.; Zhang, W. H.; Lu, N.; Zhuo, Z. W.; Zeng, X. C.; Wu, X. J.; Yang, J. L.  $CO_2$  Capture on h-BN Sheet with High Selectivity Controlled by External Electric Field. *J. Phys. Chem. C* **2015**, 119, 6912–6917.

(42) Kou, L. Z.; Du, A. J.; Cheng, C. F.; Frauenheim, T. Strain Engineering of Selective Chemical Adsorption on Monolayer MoS<sub>2</sub>. *Nanoscale* **2014**, *6*, 5156–5161.

(43) Tersoff, J.; Hamann, D. R. Theory and Application for the Scanning Tunneling Microscope. *Phys. Rev. Lett.* **1983**, *50*, 1998.